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Submicron-Sized Nb-Doped Lithium Garnet for High Ionic Conductivity Solid Electrolyte and Performance

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Description

The physico-chemical properties of the copper electrolyte significantly affect the energy consumption of the electrorefining process and the quality of the cathode product. Favorable conditions for electrorefining processes are typically achieved by keeping both the electrolyte conductivity and diffusion coefficient of Cu(II) high, while ensuring low electrolyte viscosity. In this work the conductivity of the copper electrorefining electrolyte was investigated as a function of temperature (50-70 °C) and concentrations of copper (Cu(II), 40-60 g/L), nickel (Ni(II), 0-20 g/L), arsenic (As(III), 0-30 g/L) and sulfuric acid (160-220 g/L). In total 165 different combinations of these factors were studied. The results were treated using factorial analysis, and as a result, four electrolyte conductivity models were built up. Models were constructed both with and without arsenic as the presence of As (III) appeared to cause non-linearity in some factor effects and thus impacted the conductivity in more complex ways than previously detailed in literature. In all models the combined effect of factors was shown to be minor when compared to the effect of single factors. Conductivity was shown to increase when copper, nickel and arsenic concentrations were decreased and increase with increased temperature and acidity. Moreover, the arsenic concentration was shown to decrease the level of conductivity more than previously suggested in the literature. Copper electrorefining is the most common method for producing high-purity copper. The first refinery (Pembrey Copper Works) was established in 1869 following the first patent for commercial electrorefining developed by James Elkington in 1865. Since then, the process has been developed further as the result of both research and improved industrial practices. In the copper electrorefining process, copper is dissolved from impure copper anodes into the electrolyte bath and then subsequently deposited on to cathodes as high-purity copper. In order to get an impression of how the monomers structure influences the polymers properties, we synthesized eight new ammoniumbased ionic liquid monomers, which were presented in a previous publication. The corresponding chemical structures which are labelled as [CnN(M)A,22]TFSI. Variations were applied in the polymerizable group and the length of one alkyl group at the quaternary ammonium group. For all monomeric ionic liquids the Bis(trifluoromethane)sulfonimide anion (TFSI) was used. Cn represents the length of the introduced alkyl chain at the quaternary ammonium ion. The synthesis of the monomers starts either from 2-(diethylamino)ethyl acrylate or 2-(diethylamino)ethyl methacrylate, which is indicated in the nomenclature as A,22 (for acrylate and two ethyl groups) or MA, 22 (for methacrylate and two ethyl groups), respectively. For the monomers (labelled as [Cation]Anion), we found that the structure variations have a systematic influence on the monomer's ionic conductivity. Acrylate functionalities and short side chains at the ammonium group lead to higher ionic conductivities than methacrylates and longer side chains. This observation can be explained with the higher steric demand of bulkier groups and resulting phenomena like entanglements that lead to increased internal friction, causing decreased ionic mobility.

The occurrence of molecular interactions between the functional groups of the polymer and cations

In this paper, we report the ionic conductivity and cycling stability improvement of PVDF/nanoclay as Polymer Electrolyte Membranes for LiFePO4 Batteries. This research is important because the coactions of PVP and nano-clay significantly improved the performance of the polymer electrolytes in term of the electroactive phase, the degree of crystallinity, porosity, electrolyte uptake, the C-rate discharge capacity and the ionic conductivity. To the best of our knowledge, the literature cites no reports of nano-clay PVDF PEMs prepared using PVP as poreforming agents. These nano-clay PVDF PEMs were structurally modified by PVP and showed high ionic conductivity and stable cyclability. The diffractogram of pure CS and CS doped with 20 wt % AgNt. It is clear that pure chitosan exhibits some peaks at 2θ = 15.6°, 17.4° and 21.8°. These peaks are usually characterized as a semi-crystalline polymer. The inset of the main peak of chitosan, which appeared at $2\theta = 21.8^{\circ}$, almost scarified and changed to a broad peak. Earlier study verified that the lowering of intensity and enlargement of the XRD pattern indicates an increase in amorphous phase. The occurrence of molecular interactions between the functional groups of the polymer and cations of the doping salt hinder the ordering of the crystalline phase and consequently induce an increase in amorphous fraction. The XRD pattern of pure PVA. The inset of the XRD pattern for CS:AgNt incorporated with 10 wt % PVA

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(CSPV1) and 40 wt % PVA (CSPV4) Acids are first applied in PBI PEMs for DMFCs in 1995 by Wainright et al. they applied H3PO4doped PBI as a membrane system and obtained a greatly lowered methanol crossover rate in comparison to Nafion. After 26 years, such issues are still in concern, the potential of aciddoped PBI PEMs for DMFC applications has been widely recognized. Key developments have been achieved as early as 1996, Wang et al. applied acid-doped PBI PEM to DMFC and performed real-time analysis of the methanol crossover and poisoning effect in the cathode of the fuel cell. With this, the mechanism of methanol crossover was revealed and the accompanied CO poisoning effect caused by methanol oxidation at the cathode was promoted. As the inevitable of methanol crossover and the CO gas, DMFCs working at elevated temperatures are developed to solve the problem. However, new problems appeared for acid-doped PBI membranes.

In this study, tertiary ammonium compounds of pyridine, morpholine, 3-methylimidazole, pyrrolidine, tetraalkylammonium and picolines have been alkylated with nalkyl side chains in the N-position to form organic bromide or chloride salts. In combination with some commercially available organic salts, this resulted in 38 organic quaternary ammonium substances from these six N-containing building blocks. The substances have been investigated ex situ on their properties as BCAs in HBr/Br2-electrolytes: BCA solubility, Br2 binding strength, the stability of the second heavy fused salt phase, Br2 concentration for application, electrolyte conductivity and redox potential. This study has been performed to define their applicability of BCAs to limit the vapor pressure of Br2 in aqueous highly concentrated electrolytes for a high-energy density H2/Br2-RFB with a theoretical capacity of 180 Ah L -1/196 Wh L-1 (7.7 M HBr).